УДК 538.915

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# Experimental study of irradiated few-layer graphene

**Abstract.** Few layer graphene fragments were obtained by mechanical exfoliation technique on copper grid for TEM, irradiated by 100 keV electrons and characterized by optical microscopy, scanning electron microscopy, the energy dispersive X-ray spectroscopy and Raman spectroscopy. In order to achieve higher sensitivity on surface carbon nanostructures the low-energy probing (1 keV) by the energy dispersive spectroscopy was used. Data of the energy dispersive X-ray spectroscopy allows to obtain well distinct levels of carbon and to determine a minimum step level relating to one graphene layer. Raman spectra shows apparent contribution of the D-peak, indicating radiation damage after 100 keV irradiation and conventional medium electron microscopy investigations. The weak effect of radiation damage has also been revealed after low-energy (2 keV) observation and energy dispersive X-ray measurements using 1 keV primary electron beam. *Keywords:* graphene, carbon nanostructures, mechanical exfoliation, X-ray spectroscopy, Raman spectroscopy.

## Introduction

The scientific and technological significance of graphene is linked with its good combination of both unique physical and mechanical properties. It is well known, that graphene is being very promising material for applications in modern and future nanotechnologies and can be used in many types of sensors and devices in many application areas. Therefore, graphene, in spite of its recent availability for experimental investigations [1] is an object of great interest for many researches [2-8].

Unfortunately, identification and characterization of ultrathin graphite and graphene fragments after mechanical production from highly oriented pyrolytical graphite can be performed with some limitations and difficulties. Moreover, as for graphene, its fragments are usually dispersed among more thicker few-layers and thin graphite pieces on a substrate surface. Discovering the few-layer graphene fragments in SEM or in optical microscope usually includes selecting some areas of the surface with nanoobjects by the secondary electron image contrast differences. Using of analytical methods for characterization of ultrathin graphite and graphene is not widespread till present. Taking into account that during SEM investigations radiation structural defects can be produced [3,7,9,10] in this paper we have focused on the using the combination of the low-energy SEM investigations with energy dispersive X-ray analysis and Raman spectroscopy by nondestructive searching and characterizing thin graphite and graphene fragments, arranged on a copper substrate.

### **Experiment and results**

Thin graphite and graphene pieces were arranged on a copper substrate due mechanical exfoliation of highly oriented pyrolytical graphite [1]. Two types of substrate were used: copper grid for TEM and copper polycrystal substrate after mechanical polishing. All graphite fragments were initially observed with an optical microscopy. We have supposed that large difference of atomic numbers between carbon and copper make it easier to select ultrathin carbon fragments by SEM Quanta 3D 200i dual beam system. All composition measurements were performed with the help of the EDAX Ametek system built in SEM. Raman scattering measurements were performed by NT-MDT NTegra confocal spectrometer at room temperature in backscattering geometry using the 473 nm line of a semiconductor laser. The spectrometer was calibrated using the frequency of the optical phonon peak (1332 cm<sup>-1</sup>) of mono crystalline diamond. The scattered light was dispersed and detected with cooled charge couple device detector.

The few layer graphene fragments were irradiated by TEM 100 keV electrons in Texas State University, with fluence  $2*10^{19}$  m/cm<sup>2</sup> at room temperature. The main operating regime, which was used by SEM observations: V=20 keV (the probe size was equal to 2 nm, with the beam current 0.12 nA.) by the conventional medium energy SEM operation and V=2 keV by searching graphene in low-energy mode in order to avoid radiation damage. The EDX measurements were conducted by the low-energy primary electron-beam. We have taken into account that a maximum value of ionization cross-section for a K-level with a binding energy  $E_b$  corresponds to the relation V/ $E_b \approx$ 3-4 [11]. As for carbon  $E_b=285$  eV, so the energy of the primary electron beam as large as 1 keV was selected. The number of X-rays emitted is approximately proportional to the size of the volume that is subjected to irradiation. Proportionality between graphene stack thickness and the signal of the EDX X-ray detector is observed if spot area, current and value of e-beam energy are kept constant. During all EDX measurements parameters were used as follows: spot of 2 nm diameter, 1 keV e-beam energy and the beam current 0.12 nA. The composition of the copper surface free of graphite flaks, determined by the EDX technique at the energy of the electron beam 2 keV was as follows: Cu - 46%, C - 44%, O - 10% (at%). The level of the background surface impurity of carbon was nearly constant on the surface with very low variations ( $\approx 2\%$ ).

Figure 1 presents typical image of areas with carbon layers on surface obtained by using lowenergy primary electron beam (2 keV). One can see several areas which are well distinguished on the contrast. Figure 2 presents EDX - intensities of carbon at marked areas after background subtraction. The background level, measured in the same arbitrary units was equal to  $14.0 \pm 0.3$ .



**Figure 1** – Optical imagies of graphite flakes with the marked areas



Figure 2 – SEM imagies of graphite flakes with the marked areas

EDX signal was recorded in 5 different points for every fragment. The accuracy of the data presented in Figure 2 is equal to 5%. One can see that all histograms in Figure 2 show obviously discrete structure and are approximately multiples of 5 a.u., and moreover, the minimum intensity level which corresponds to the layer with a minimum contrast (the fragment 4) is also equal to  $5 \pm 0.1$  a.u. It would be reasonably to suppose, that this minimum intensity level corresponds to single graphene layer.



Figure 3 – The carbon EDX intensities from the marked areas presented in Fig.2

In order to confirm that we deal with very thin layers of graphite, the additional measurement of the EDX intensities by different angles  $\theta$  between the normal to the surface investigated and the electron beam direction were performed (Fig.3). One can see, that data presented are in a fine agreement with

a "I<sub>0</sub>/cos  $\theta$ " - dependence, where I<sub>0</sub> is the intensity of carbon at  $\theta = 0$ . It confirms that in our case the thikness of the fragment investigated is less than the depth of X-rays generation from carbon K-level (285 eV) for 1 keV electron beam, which is less than nearly 1 nm [11].



The measurements were performed on the area 2.

Figure 4 – A typical dependence of the EDX intensity on tilt angle

Figure 5 presents typical Raman spectra of the fragments investigated. It is known [6,10], that graphite mono-crystal has a single Raman active mode, which is zone center mode at nearly 1580 cm<sup>-1</sup> labeled "G" for graphite. Disordered graphite has a second mode at around 1350 cm<sup>-1</sup> labeled "D" for "disorder". It corresponds to the breathing vibrations of rings at the graphitic cluster. This mode is forbidden in perfect graphite and only becomes active in the presence of disorder. Fig.4 allows to compare effect of low- and medium energy electron beams. In pristine fragment one can see only the

distinct G peak near 1585 cm<sup>-1</sup>, that indicates the high quality of crystalline structure. After using the low-energy probe (2 keV SEM- observation and 1 keV EDX-measurements), the Raman spectrum reveals G- and weak D-peak, which indicates the low contribution of a defect structure. One can see, that after using the medium energy regime of observation (20 keV), D-peak is obviously much more larger. This result is in a good agreement with data, recently reported in [10] about a radiation damage observed in graphene structure under SEM observations.



Figure 5 - Raman spectra of sample before and after energy 100 keV beam irradiation

Figure 5 shows a proposed typical picture of three layer graphene with bridge-like radiation defects which link together graphene sheets. One can see that the effective distance between graphene layers after irradiation becomes smaller than in initial state. This computer model explains essential changes in Raman spectra after irradiation. In more detailed results of computer simulation of bridgelike radiation defects were published recently in [12,13].

## Conclusion

In conclusion, we have irradiated by fast on the copper surface thin graphite and grapheme fragments obtained by mechanical exfoliation and used low-energy scanning electron microscopy, energy dispersive X-ray and the Raman spectroscopy to characterize carbon nanostructures. Well distinguished changes in Raman spectra (intensities and positions of G,D and 2D peaks) were observed. Discrete steps between carbon peak intensities, obtained by the low-energy dispersive X-ray spectroscopy was interpreted as indicating to variation of the depth equal to one or a few graphene layers.

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## И.А. Цыганов, Н.Р. Гусейнов, Р.Р. Немкаева, Б.А. Борисов, А.М. Ильин Бірнеше қабаты бар графеннің сәулелендірілген бөліктерін тәжірибе жүзінде зерттеу

Графеннің бірнеше қабаттарының бөліктері ТЕМ үшін мыстан жасалған тордың бетінде механикалық жаңқалау әдісі арқылы алынған болатын. Кейін үлгілер энергиясы 100 кэВ жететін электрондармен сәулелендіріліп, оптикалық және сканирлеуші электрондық микроскопия, энергодисперсиялы рентгендік, рамандық спектроскопия әдістерімен зерттелген болатын. Сезімталдығын жоғарылату мақсатында көміртекті наноқұрылымдардың бет-бедеріне энергиясы төмен энергодисперсиялық спектроскопия қолданылған болатын. Энергодисперсиялы рентгендік спектроскопия арқылы алынған нәтижелердің көмегімен жақсы ажыратылатын көміртек деңгейлерін алуға және графеннің бірлік қабатына сәйкес келетін минимал деңгейін анықтауға мүмкіндік берді. Раман спектрінде D-пикінің пайда болуы соған сәйкес электрондармен (100 кэВ) және электрондық микроскопия арқылы сәулелендіру процестерінен кейін радиациялық ақаулардың пайда болуы туралы дәлелдейді. Радиациялық ақаулардың әлсіз эффектісі энергиясы 1 кэВ болатын екінші реттік электрондар ағынын пайдалана отырып, төменгі энергиялық (2 кэВ) энергодисперсиялы рентгендік өлшемдерден кейін байқалды.

*Түйін сөздер:* графен, көміртекті наноқұрылымдар, механикалық ұсақтап жару, рентген спектроскопиясы, Раман спектроскопиясы.

# И.А. Цыганов, Н.Р. Гусейнов, Р.Р. Немкаева, Б.А. Борисов, А.М. Ильин Экспериментальное изучение облученного многослойного графена

Фрагменты нескольких слоев графена были получены методом механического отщепления на медных сетках для TEM, далее образцы были облучены электронами 100 кэВ и проанализированы с помощью оптической и сканирующей электронной микроскопии, энергодисперсионной рентгеновской и рамановской спектроскопии. Для обеспечения более высокой чувствительности на поверхности углеродных наноструктур была применена низкоэнергетичная энергодисперсионная спектроскопия. Данные энергедисперсионной рентгеновской спектроскопии позволили получить хорошо различимые уровни углерода и определить минимальный уровень соответствующий единичному слою графена. Рамановский спектр показывает появление D-пика, свидетельствуя о радиационном повреждении после облучения электронами (100 кэВ) и после обычного изучения с помощью электронной микроскопии. Слабый эффект радиационных повреждений также наблюдался после низкоэнергетичного (2 кэВ) наблюдения и энергодисперсионного рентгеновского измерения с использованием первичного электронного пучка энергией в 1 кэВ.

*Ключевые слова:* графен, углеродные наноструктуры, механическое отщепление, рентгеновская спектроскопия, рамановская спектроскопия.